# Thermal Degradation of Polymers. XII. Synthesis, Polymerization and Thermal Characterization of Polymers of *m*-N,N-Dimethylaminostyrene in Nitrogen

# S. OPREA\* and R. H. STILL, Department of Polymer and Fibre Science, UMIST, Manchester, England

### **Synopsis**

The monomer m-N,N-dimethylaminostyrene has been synthesized from acetophenone and homopolymerized and copolymerized with styrene, using 2,2'-azobisisobutyronitrile as free-radical initiator. The polymers have been characterized by thermal analysis (DTA, DSC, and TG). Glass transition temperatures have been measured and activation energies for degradation in nitrogen have been established by a variety of methods for comparative purposes.

### INTRODUCTION

In parts X and XI of this series,<sup>1,2</sup> we discussed the effect of molecular weight and copolymer composition on the thermal behavior of poly(p-N,N-dimethylaminostyrene) (PDAS) homopolymers and copolymers<sup>1</sup> and poly(p-N,N-diethylaminostyrene) (PDEAS) homopolymers and copolymers<sup>2</sup> in nitrogen.

We now report studies on the synthesis, polymerization, and copolymerization of m-N,N-dimethylaminostyrene polymers (PmDAS), and their characterization by thermal analytical techniques [thermogravimetry (TG), differential thermal analysis (DTA), and differential scanning calorimetry (DSC)].

## **EXPERIMENTAL**

# Materials

The monomer was synthesized from acetophenone by the following route:

\* Present address: Department of Natural and Synthetic Macromolecular Compounds, Polytechnic Institute of Iasi, Iasi, Romania.

© 1976 by John Wiley & Sons, Inc.



Acetophenone was converted to m-nitroacetophenone (I) by a method previously reported.<sup>3</sup> Compound (I) was converted to m-aminoacetophenone by modification of a previously reported method for the reduction of *m*-nitrostyrene.<sup>3</sup> Stannous chloride dihydrate (206 g) and concentrated hydrochloric acid (133 ml) were stirred together for 30 min in a 1-liter, three-necked flask equipped with a reflux condenser, a dropping funnel, and a thermometer. The reaction vessel was maintained at 70-80°C, and a green-yellow solution was obtained to which was added over a period of 30 min a solution of m-nitroacetophenone mp 77-78°C (33 g) in ethanol (50 ml). After addition was complete, the solution was boiled under reflux for 1 hr. The reaction mixture was then cooled and added dropwise to a vigorously stirred ice-cold solution of sodium hydroxide (266 g) in water (700 ml). The precipitated product was filtered and extracted with chloroform  $(4 \times 75 \text{ ml})$ . The chloroform extract was dried over anhydrous sodium sulfate and the chloroform removed by distillation. It gave on recrystallization from industrial methylated spirit maminoacetophenone mp 94-95°C in 85% yield.

The conversion of m-aminoacetophenone to m-dimethylaminoacetophenone was effected by a modification of Komogawas procedure.<sup>5</sup> m-Aminoacetophenone (13.6 g) and a 7% solution of sodium carbonate (500 ml) were stirred together in a 1-liter, three-necked flask equipped with a reflux condenser, a dropping funnel, a thermometer, and a magnetic stirrer. Dimethyl sulfate (32 ml) was added dropwise to the stirred solution over a period of 1 The temperature of the reaction media was then raised to 60°C and hr. maintained at this temperature for 1 hr. The unreacted dimethyl sulfate was then decomposed by the addition of solid sodium hydroxide (20 g). m-N,N-Dimethylaminoacetophenone was then extracted from the reaction medium with ether  $(4 \times 75 \text{ ml})$ . The combined ethereal extract was dried over anhydrous sodium sulfate and the ether removed by rotary film evaporation. The resultant product was boiled under reflux for 3 hr with a threefold excess of acetic anhydride to remove the monomethyl derivative. The cooled solution was dissolved in ether and extracted with 2N hydrochloric acid. The acid extract was neutralized with 2N sodium hydroxide and m-N,N-dimethylaminoacetophenone extracted with ether. The ethereal extract was dried over anhydrous sodium sulfate and the ether removed by rotary film evaporation.

*m*-N,N-Dimethylaminoacetophenone was obtained by distillation at reduced pressure. It had bp  $110-112^{\circ}$ C/0.5 mm Hg (lit.<sup>5</sup> 98°/4 mm); yield 63%.

*m*-N,N-Dimethylaminoacetophenone (III) was converted into  $\alpha$ -methyl*m*-N,N-dimethylaminobenzyl alcohol (IV) by reduction with aluminum isopropoxide by the following method:

Aluminum foil (4.72 g), mercuric chloride (0.055 g), and dry isopropanol (194 ml) were placed in a 500-ml flask equipped with a reflux condenser, a dropping funnel, and a thermometer. The isopropanol was boiled under reflux until the aluminum had dissolved (6 hr). m-N,N-Dimethylaminoacetophenone (55 g) was then added to the solution and the reaction mixture was boiled under reflux for a further 2 hr. The excess of isopropanol and the acetone formed in the reaction were then removed by distillation at slightly reduced pressure. The residue was then treated with 2N sodium hydroxide and then extracted with ether. The ethereal extract was dried over anhydrous sodium sulfate and the ether removed and the product obtained by fractional distillation at  $101-103^{\circ}C/0.09 \text{ mm Hg}$ ,  $N_D^{25}$  1.5555 in 64% yield (lit.<sup>6</sup> bp 126-128°C/2 mm Hg,  $N_D^{22}$  1.5620).

The dehydration of  $\alpha$ -methyl-*m*-N,N-dimethylaminobenzyl alcohol was effected in a similar manner to that of Mannecke and Kossmehl.<sup>6</sup> Compound IV (5 g) was added dropwise to activated alumina (16 g) maintained at 280–300°C in a pear-shaped distillation flask, and the products of reaction allowed to distil out of the reaction vessel under nitrogen at a pressure of 0.1–0.2 mm Hg. The distillate was condensed in a receiver cooled in an ice–salt bath. The colorless liquid so obtained was dried over anhydrous sodium sulfate and redistilled in nitrogen under vacuum in the presence of diphenyl picryl hydrazyl. The main fraction obtained in 74% yield had bp 57–59°C/0.025 mm Hg,  $N_D^{22}$  1.5872 (lit.<sup>5</sup> bp 105°C/2 mm; lit.<sup>7</sup>  $N_D^{22}$  1.5863).

ANAL. Found: C, 81.30%; H, 8.70%; N, 9.40%. Calcd for  $C_{12}$  H $_{13}$  N: C, 81.63%; H, 8.84%; N, 9.52%.

The monomer was homopolymerized in bulk under nitrogen at 70°C in sealed soda-glass test tubes using  $\alpha, \alpha'$ -azobisisobutyronitrile (AZBN) as initiator at concentrations between 0.2% and 3.2% on monomer weight. This yielded poly(*m*-N,N-dimethylaminostyrenes) (PmDAS) 1-4 of atactic stereochemical configuration and different molecular weights. Purification prior to characterization was effected in the following manner. The solid rods of polymer were dissolved in AnalaR toluene to give 5% w/v solutions which were added dropwise to excess mechanically stirred isopropanol. The resulting fibrous polymer was filtered at the pump through sintered glass crucibles and washed with isopropanol and dried at 20°C to constant weight at 2 mm Hg (three to four days). Three representative samples, PmDAS 1-3, were subjected to combustion analysis.

ANAL. Found for PmDAS 1: C, 81.50%; H, 9.10%; N, 9.30%; PmDAS 2: C, 81.80%; H, 8.90%; N, 9.40%; PmDAS 3: C, 81.70%; H, 9.00%; N, 9.50%. Calcd for  $C_{10}H_{13}N$ : C, 81.63%; H, 8.84%; N, 9.52%.

PmDAS samples were characterized by intrinsic viscosity determination in toluene at 25° using a Ubbelohde No. 1 dilution viscometer. The data presented in Table I show the polymers to have sensibly different molecular weights.

PmDAS sample no.	AZBN, %	Intrinsic viscosity $[\eta]_{0}$
1	0.18	0.83
$\overline{2}$	1.11	0.35
3	1.77	0.32
4	3.26	0.15

 TABLE I

 Intrinsic Viscosities for PmDAS Homopolymer

TABLE II S:m-DAS Copolymer Characterization Data

Sample no.	AZBN, %	Approx. molar ratio S:mDAS	Molar ratio from combustion S:mDAS analysis	Intrinsic viscosity $[\eta]_0$
5	1.16	2.5:1	2.43:1	0.42
6	0.99	1.5:1	1.56:1	0.48
7	1.09	0.5:1	0.56:1	0.55

Copolymerization of styrene (S) and m-N,N-dimethylaminostyrene (m-DAS) in various molar ratios under similar conditions with an initiator concentration of  $\simeq 1.0\%$  on total monomer weight yielded copolymers 5, 6, and 7 after 6 hr as solid glasses. Purification and characterization by viscometry was effected as previously described for the homopolymers. Combustion analysis was performed on all copolymer samples.

ANAL. Found for copolymer 5: C, 88.60%; H, 8.30%; N, 3.40%. Calcd for  $C_8H_8:C_{10}H_{13}N = 2.43:1$ : C, 88.39%; H, 8.11%; N, 3.49%. Found for copolymer 6: C, 87.00%; H, 8.10%; N, 3.49%. Calcd for  $C_8H_8:C_{10}H_{13}N = 1.56:1$ : C, 87.23%; H, 8.23%; N, 4.52%. Found for copolymer 7: C, 84.70%; H, 8.70%; N, 6.80%. Calcd for  $C_8H_8:C_{10}H_{13}N = 0.56:1$ : C, 84.71%; H, 8.50%; N, 6.77%.

The data presented in Table II show the copolymers to have similar molecular weights.

Polystyrene samples 8 and 9 were prepared and characterized as previously reported<sup>2</sup> (part XI of this series) for samples PS10 and 11.

## **Apparatus and Procedure**

Thermal analysis studies using differential thermal analysis (DTA), differential scanning calorimetry (DSC), and dynamic and isothermal thermogravimetry (TG) were made as previously reported in part XI.<sup>2</sup>

## **RESULTS AND DISCUSSION**

The monomer *m*-DAS has been synthesized from acetophenone by modification of the known route, and polymerized using a nonoxidizing radical source (AZBN) to yield homopolymers 1-4 of different molecular weights and copolymers with styrene (5–7) having different compositions.

Reference to Tables I and III shows the normal initiator concentration/ molecular weight relationship to be obeyed for the homopolymer systems.

The effect of molecular weight on the glass transition temperature of the

Glass Transition Temperatures for FmDAS homopolymers				
Polymer	sample no.	$[\eta]_{\mathfrak{o}}$	$\overline{M_n} \times 10^{-4}$	$T_g$ , ° C
PmDAS	1	0.83	38.65	86
PmDAS	2	0.35	8.269	83
PmDAS	3	0.32	7.047	82
PmDAS	4	0.15	1.821	80
PS	10	0.28	4.980a	96
PS	11	1.23	38.900a	100

TABLE III lass Transition Temperatures for PmDAS homopolymers

<sup>a</sup> Calculated using  $[\eta]_0 = KM^{\alpha}$ , where  $\alpha = 0.72$  and  $K = 1.16 \times 10^{-4}$ .

		TA	BLE IV	
$T_g$	Values for	Various	Substituted	Polystyrenes*

Polymer	Position of substitution	
	meta	para
Poly(methylstyrene)	82	105
Poly(chlorostyrene)	90	128
Poly(N,N-dimethylaminostyrene) <sup>a</sup>	87	146

a Reference 1 and this paper.

PmDAS homopolymers is shown in Table III, together with those for representative PS samples. The results for PS have been discussed in a previous paper.<sup>2</sup>

The molecular weight data reported for PmDAS was calculated using the constants obtained experimentally for the related poly(p-N,N-dimethylaminostyrene) homopolymers (PDAS) in toluene, namely,  $\alpha = 0.56$  and  $K = 6.17 \times 10^{-4}$ .

It is apparent that the effect of molecular weight on  $T_g$  for PmDAS is of a similar order to that observed for poly(p-N,N-diethylaminostyrene) (PDEAS) reported previously,<sup>2</sup> viz., 100°C at  $\bar{M}_n$  24,900, 105°C at  $\bar{M}_n$  183,600. The magnitude of the  $T_g$  and molecular weight effects observed for the present polymers contrasts markedly with that shown by p-N,N-dimethylaminostyrene (p-DAS) homopolymers reported in an earlier paper.<sup>1</sup> The  $T_g$  of PDAS ranges from 109°C at  $\bar{M}_n$  12,200, rising to a  $T_{g^{\infty}}$  value of 146°C.

The overall reduction in the value of the  $T_g$  of PmDAS compared with PDAS is similar to that observed by Barb<sup>8</sup> for methyl- and chloro-substituted polystyrenes, as shown in Table IV.

Barb ascribed the reduction of the  $T_g$  in the case of meta substituents by reference to atomic models, to "loosening" of the structure without offering the opportunity of steric interference of the substituent with segments of other chains, which he showed to occur with the corresponding para substituents resulting in chain stiffening.

The increase in  $T_g$  for chloro-substituted styrenes over that observed for the similarly sized methyl-substituted styrene and for PS was ascribed to the additional dipolar effect in the chlorostyrene system due to the electronic effects (-I + M) of the substituent chlorine atom.

Similar effects to those suggested by Barb may account for the increase in

Copolymer sample no.	Molar ratio <sup>a</sup> S:m-DAS	[η] <sub>0</sub>	T <sub>g</sub> , °C
5	2.43:1	0.42	94
6	1.56:1	0.48	92
7	0.56:1	0.55	84

TABLE V  $T_g$  for S:m-DAS Copolymers

<sup>a</sup> From combustion analysis.

the  $T_g$  of PDAS<sup>1</sup> compared with PS and PmDAS. The reduction in  $T_g$  for PmDAS compared with PS and PDAS arises as a result of the effects of the nonsymmetrical, pendent, substituted phenyl nucleus on chain "loosening" interchain steric hindrance and dipolar interaction.

The smaller effect of molecular weight on  $T_g$  in the PmDAS system compared with PDAS may be ascribed to the effect of the nonsymmetrical substituted phenyl group on the overall "free volume" of the polymer, which is relatively unaffected by increasing the number of chain ends in the system. If the data presented in Table III are plotted according to the Flory-Fox equation,<sup>9</sup> the curve shown in Figure 1 is obtained. This shows the equation to hold for this system only at high molecular weights if the assumptions made in deriving the molecular weight data are tenable.<sup>2</sup> Computation of the free volume of the chain end from the initial slope of the curve by the method previously described for PDAS<sup>1</sup> leads to a value of 79 Å<sup>3</sup>.

Table V shows the glass transition temperatures for the copolymer systems. In the range studied, it is clear that as the molar percentage of S is in-



Fig. 1. Flory-Fox plot for PmDAS homopolymers.

Polymor	Ten	nperature, °C, for s	specified % weight	loss
sample	()a	10	50b	75
PmDAS 1	338	393	426	439
PmDAS 2	343	402	430	446
PmDAS 3	348	402	431	447
PmDAS 4	340	402	<b>434</b>	447
PS 10	314	394	421	431
PS 11	357	402	422	430

TABLE VIComparative Thermal Stabilities of PmDAS and PS as Assessedby TG in Dynamic Nitrogen

a PDT (procedural decomposition temperature).

b 50% DT (50% decomposition temperature).

creased, the  $T_g$  increases. This behavior is again different from that reported previously for styrene/p-DAS copolymers<sup>1</sup> and suggests a reduction in the chain "loosening" effect of the *m*-N,N-dimethylamino substituent and an increase in van der Waals interaction as the system is progressively diluted with substituted phenyl units. In no case was more than one  $T_g$  observed, confirming the random nature of these copolymer systems.



Fig. 2. DSC curves in nitrogen for PmDAS and PS: (1) PS sample weight 4.05 mg; (2) PmDAS 1, sample weight 4.50 mg.

Copolymer S:m-DAS sample no. ratio <sup>a</sup>	Sum DAS	Tempe	erature, °C, for s	pecified % weig	t loss
	0ь	10	50c	75	
5	2.43:1	338	389	421	435
6	1.56:1	338	393	424	438
7	0.56:1	330	385	422	438

TABLE VII Comparative Thermal Stabilities of S:m-DAS Copolymers Assessed by TG in Dynamic Nitrogen

<sup>a</sup> Evaluated by combustion analysis.

b PDT.

c 50% DT.

-

TABLE VIII Heats of Degradation in Dynamic Nitrogen

## **Thermal Stability Studies**

The polymers were subjected to thermal degradation in nitrogen to assess the effect of the *m*-N,N-dimethylamino substituent on the degradation behavior and the thermal stability as compared to PS. The behavior of PmDAS homopolymers and the two PS samples is compared in Table VI. The data show that within experimental error for PmDAS samples, molecular weight had no effect on thermogram shape or the derived parameters such as PDT or 50% DT values. Isothermal studies similarly revealed no significant differences between PmDAS 1–4 with  $W_{30}$  at 400°C being 93%, compared with a  $W_{30}$  value for PS of 94%. The data for PS were discussed in the previous paper.<sup>2</sup>

The values obtained for PS and PmDAS and  $PDEAS^2$  indicate that these systems have similar stabilities as assessed by dynamic and isothermal methods. Further to this, the DSC traces (Fig. 2) show marked similarity with a corrected peak minimum of 422°C for PS and 425<sup>b</sup>C for PmDAS compared with PDEAS<sup>2</sup> 414°C.

The behavior of PmDAS contrasts with that reported for PDAS<sup>1</sup> (PDT, 350°C; 50% DT, 414°C;  $W_{30}$ , 47.5%), suggesting that a different mechanism of degradation operates in the two systems. Vacuum pyrolysis of PmDAS and product analysis studies are currently in progress and will be the subject of a future publication.

Comparative studies on the S:m-DAS copolymers are shown in Table VII. The data indicate that the stability is independent of copolymer composition in the range studied.

	Activation energy, kcal/mole		
Polymer	Method I <sup>a</sup>	Method II <sup>b</sup>	
PmDAS 1	51.4	52.6	
S:m-DAS 6	50.9	49.8	
PS	46.6	48.7	

TABLE IX Kinetic Parameters Evaluated by Thermogravimetry

<sup>a</sup> At 0.4 fractional conversion.

<sup>b</sup> Assuming a first order reaction.

	Real	eich metho	d	Friedman me	ethod
Compound	E, kcal/mole	n	Temp, °C	<i>E</i> , kcal/mole	n
PmDAS 1	54.5	1.33	381-461	47.0	0.91
S:mDAS 6	53.1	1.19	381 - 454	50.8	1.30
PS	51.8	0.97	381 - 450	57.8	1.34

TABLE XKinetic Parameters Evaluated by DSC

Heats of reaction data from DSC studies are shown in Table VIII which show an increase in  $\Delta H$  as the styrene content of the system increases. As can be seen, the molecular weight of the PmDAS samples has little effect on the value of  $\Delta H$  obtained.

## **Kinetic Studies**

Comparative studies were made using TG and DSC. Activation energies were derived from dynamic TG data by the method of MacCallum and Tanner<sup>10</sup> (method 1) and from isothermal data (method II), and the results are shown in Table IX. Similar studies were made using DSC by the methods of Friedman<sup>11</sup> and Reich<sup>12</sup>; the results obtained are shown in Table X. The results obtained are subject to the limitations previously discussed for *p*-N,Ndiethylaminostyrene polymers.<sup>2</sup> However, it is apparent that these systems have the same general order of stability, with PS being slightly less stable.

One of the authors (S.O.) thanks the British Council for a Research Fellowship.

#### References

1. R. H. Still and A. Whitehead, J. Appl. Polym. Sci., 20, 627 (1976).

2. R. H. Still and S. Oprea, J. Appl. Polym. Sci., 20, 639 (1976).

3. C. L. Arcus and R. H. Still, J. Chem. Soc., 4340 (1964).

4. Dictionary of Organic Compounds, 4th ed., J. R. A. Pollock and R. Stevens, Eds., Eyre and Spottiswoode Publishers, London, 1965.

5. H. Komagawa, M. Kato, and H. Sugijawa, J. Polym. Sci. A-1, 6, 2967 (1968).

6. G. Mannecke and G. Kossmehl, Makromol. Chem. 80, 22 (1964).

7. G. E. Eigenmann, Dissertation, University of Missouri, 1957.

8. W. G. Barb, J. Polym. Sci., 37, 515 (1959).

9. T. G. Fox and P. J. Flory, J. Appl. Phys., 21, 587 (1950).

10. J. R. MacCallum and J. Tanner, Eur. Polym. J., 6, 907 (1970).

11. H. L. Friedman, J. Polym. Sci., B7, 41 (1969).

12. L. Reich Makromol. Chem., 123, 42 (1969).

Received May 30, 1975 Revised June 16, 1975